

# Hierarchical Mn<sub>2</sub>O<sub>3</sub> Hollow Microspheres as Anode Material of Lithium Ion Battery and Its Conversion Reaction Mechanism Investigated by XANES

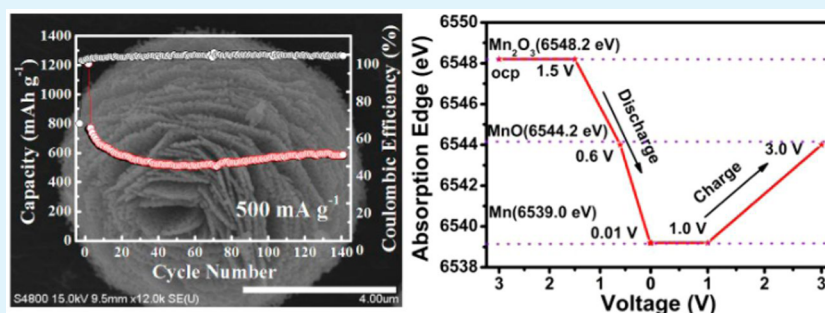
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## S Supporting Information



**ABSTRACT:** Hierarchical Mn<sub>2</sub>O<sub>3</sub> hollow microspheres of diameter about 6–10  $\mu\text{m}$  were synthesized by solvent-thermal method. When serving as anode materials of LIBs, the hierarchical Mn<sub>2</sub>O<sub>3</sub> hollow microspheres could deliver a reversible capacity of 580 mAh g<sup>-1</sup> at 500 mA g<sup>-1</sup> after 140 cycles, and a specific capacity of 422 mAh g<sup>-1</sup> at a current density as high as 1600 mA g<sup>-1</sup>, demonstrating a good rate capability. Ex situ X-ray absorption near edge structure (XANES) spectrum reveals that, for the first time, the pristine Mn<sub>2</sub>O<sub>3</sub> was reduced to metallic Mn when it discharged to 0.01 V, and oxidized to MnO as it charged to 3 V in the first cycle. Furthermore, the XANES data demonstrated also that the average valence of Mn in the sample at charged state has decreased slowly with cycling number, which signifies an incomplete lithiation process and interprets the capacity loss of the Mn<sub>2</sub>O<sub>3</sub> during cycling.

**KEYWORDS:** Mn<sub>2</sub>O<sub>3</sub>, hierarchical hollow microspheres, lithium ion battery, conversion reaction mechanism, XANES

## 1. INTRODUCTION

Lithium ion batteries (LIBs) are widely used in mobile phones, portable electronic devices, and notebook computers.<sup>1,2</sup> However, since the graphite anode material of commercial LIBs is approaching its inherent limit in performance, the LIBs could hardly fulfill the growing demands of fast development of electric vehicles (EVs), hybrid EVs, and plug-in hybrid EVs in terms of energy density, power, and safety. Transition metal oxides such as iron oxide,<sup>3,4</sup> cobalt oxide,<sup>5–7</sup> nickel oxide,<sup>8</sup> and manganese oxide<sup>9,10</sup> have a higher energy density than graphite and have been widely studied as alternative anode materials of LIBs. Manganese oxides, as one kind of transition metal oxides, have a high theoretical specific capacity, can operate at a low conversion potential, and are abundant with low cost and environmental benignity. They are therefore appropriate for the electrode materials of LIBs.<sup>11</sup> Among the manganese oxides, Mn<sub>2</sub>O<sub>3</sub> has a theoretical capacity as high as 1018 mAh g<sup>-1</sup>, while it has a low electronic conductivity and poor cycle ability

due to large volume expansion during charge–discharge, which hampered its application as anode in rechargeable LIBs.<sup>12,13</sup> Many studies have recently focused on the structure-controlled synthesis of Mn<sub>2</sub>O<sub>3</sub> in order to enhance their electrochemical performances.<sup>14</sup> Hu et al. prepared Mn<sub>2</sub>O<sub>3</sub> microspheres assembled from porous nanosheets.<sup>15</sup> The Mn<sub>2</sub>O<sub>3</sub> microspheres could deliver a reversible capacity of 748 mAh g<sup>-1</sup> at 50 mA g<sup>-1</sup> over 45 cycles. Chen et al. synthesized Mn<sub>2</sub>O<sub>3</sub> microstructured spheres and polyhedrons,<sup>16</sup> and demonstrated that the Mn<sub>2</sub>O<sub>3</sub> samples could deliver a capacity of only 190 mAh g<sup>-1</sup> after 15 cycles.

The cycle performance of the Mn<sub>2</sub>O<sub>3</sub> electrodes is strongly related to the tremendous volume change in conversion reaction. One of the widely adopted strategies, in general, to

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improve the cycling performance is to synthesize the  $\text{Mn}_2\text{O}_3$  particles to the nanometer scale.<sup>17–21</sup> However, nanoparticles in electrodes easily self-aggregate under a relatively high current density owing to their high surface energy, which results in reducing the effective contact areas of electrode materials and electrolyte and leads consequently to gradual decrease in capacity.<sup>22</sup> To further improve the cycle performance of the transition metal oxide electrodes, hierarchical micro/nanostructure is regarded as an optimized design to keep the electrodes stable in charge/discharge processes,<sup>23–25</sup> because the hierarchical micro/nanostructure can inherit the advantages from both the nanoparticles and the micro-sized aggregation and offer thus a high structure stability in mechanical strength. Moreover, constructing the anode materials with hollow structure could be also an effective strategy to buffer the large volume variation.<sup>26–29</sup>

To the best of our knowledge, despite many efforts focusing on how to improve the electrochemical performance of  $\text{Mn}_2\text{O}_3$  electrodes were reported, the conversion reaction mechanism of  $\text{Mn}_2\text{O}_3$  electrodes has not been well-defined. Many results support that the conversion reaction of  $\text{Mn}_2\text{O}_3$  electrodes undergoes a transformation between  $\text{MnO}_x$  ( $0 < x < 1$ ) and  $\text{Mn}$ .<sup>30–33</sup> By analyzing the electrochemical results, on the contrary, Deng et al. speculated that porous  $\text{Mn}_2\text{O}_3$  microspheres could be oxidized to  $\text{Mn}^{3+}$  after charging to 3 V.<sup>34</sup> It is still a challenge so far to make clear the conversion reaction mechanism and capacity fading of the  $\text{Mn}_2\text{O}_3$  anode. X-ray absorption near edge structure (XANES) spectroscopy is an excellent technique to characterize the valence state of element in compounds,<sup>35,36</sup> and it was applied in this study to investigate the conversion reaction mechanism and the origin of capacity fading of  $\text{Mn}_2\text{O}_3$  electrodes.

Some metal oxides anode materials of LIBs synthesized from metal-EG precursors have been reported lately.<sup>15,37</sup> The annealing of metal-EG precursors was demonstrated as an efficient method to obtain metal oxide materials with special morphology. In the current paper, hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres were synthesized by an improved solvent-thermal method using polyvinylpyrrolidone (PVP K-30) in the synthesis process and followed by calcination. The hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres could deliver a reversible capacity of  $580 \text{ mAh g}^{-1}$  at  $500 \text{ mA g}^{-1}$  after 140 cycles, and with a coulombic efficiency of 99.7%. A high capacity of  $422 \text{ mAh g}^{-1}$  could be still obtained with an extremely high current density of  $1600 \text{ mA g}^{-1}$ . The hierarchical hollow structure  $\text{Mn}_2\text{O}_3$  materials resulted in flexibility for volume change, and thus ensured a structural stability during charge–discharge cycling. Moreover, the XANES studies have revealed, for the first time, the conversion reaction process and the origin of capacity fading of the  $\text{Mn}_2\text{O}_3$  electrodes.

## 2. EXPERIMENTAL SECTION

**2.1. Materials Synthesis.** The porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres were synthesized through annealing Mn-ethylene glycol (EG) precursors under air atmosphere at  $750 \text{ }^\circ\text{C}$  for 10 h. Next, 0.6 g of manganese acetate tetrahydrate and 0.1 g of polyvinylpyrrolidone (PVP K-30) were mixed with 75 mL of EG. The as-obtained mixture was stirred for 30 min and then transferred into a 100 mL Teflon liner. The autoclave was sealed and maintained at  $180 \text{ }^\circ\text{C}$  for 24 h, and finally cooled down to room temperature. The precipitate was collected by filtration, washed alternately with ethanol five times, and dried in a vacuum oven at  $80 \text{ }^\circ\text{C}$  for 10 h. The precursors were further annealed at  $750 \text{ }^\circ\text{C}$  for 10 h in air with a temperature changing rate of  $5 \text{ }^\circ\text{C min}^{-1}$  to yield the final products. The synthesis of  $\text{Mn}_2\text{O}_3$

microsheets were under the same experiment conditions as those mentioned above but without PVP K-30.

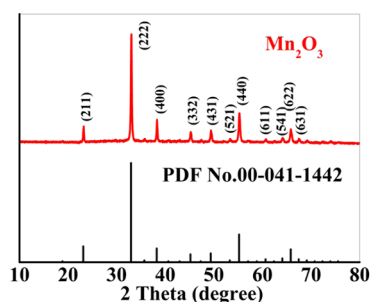
**2.2. Materials Characterization.** The phase purity and crystalline structure of the as-synthesized  $\text{Mn}_2\text{O}_3$  material was identified by powder X-ray diffraction (Philips X'Pert Pro Super X-ray diffractometer, Cu  $K\alpha$ , radiation operating at  $40 \text{ kV} \times 40 \text{ mA}$ ). Morphology of  $\text{Mn}_2\text{O}_3$  microspheres was investigated by scanning electron microscopy (SEM, HITACHI S-4800). Transmission electron microscopy (TEM, FEI Tecnai-F30FEG) was used to analyze the structure of the  $\text{Mn}_2\text{O}_3$  sample. The nitrogen adsorption/desorption isotherms were measured with a Micromeritics Tristar 3000 system, and the specific surface area was calculated by the Brunauer–Emmett–Teller (BET) method. Fourier transform infrared (FT-IR) analysis was carried out using pressed KBr disks in the range  $4000\text{--}400 \text{ cm}^{-1}$  using a Nicolet 330 spectrometer. Ex situ measurements of X-ray absorption fine structure (XAFS) were carried out at the beamline (BL14W1) at the Shanghai Synchrotron Radiation Facility (SSRF). In the Ar filled glovebox, thin-film  $\text{Mn}_2\text{O}_3$  electrodes at different cutoff voltage and cycling numbers were obtained by disassembling the cells and washing with dimethyl carbonate (DMC), and then the electrodes were reassembled back into 2025 coin cells on which a hole of 2 mm in diameter was drilled for the XAFS studies.

**2.3. Electrochemical Measurements.**  $\text{Mn}_2\text{O}_3$ , carbon (acetylene black), and the binder of LA-133 (14.96 wt %) were mixed with a weight ratio of 70:20:10 in deionized water. The obtained slurry was pasted on Cu foil and dried in a vacuum oven at  $100 \text{ }^\circ\text{C}$  overnight. The average weight of the working electrode materials was approximately  $3 \text{ mg cm}^{-2}$ . The electrochemical performances of  $\text{Mn}_2\text{O}_3$  electrodes were evaluated using CR2025 coin cells assembled in a glovebox filled with argon. The coin cell consisted of a working electrode, Li metal foil as both counter electrode and reference electrode, a Celgard 2400 separator, a liquid electrolyte mixture (TINCl, Guangzhou) that contains  $1 \text{ mol L}^{-1}$  LiPF<sub>6</sub>, and a solvent mixture of EC/DEC/DMC in a volume ratio of 1:1:1 with 2 wt % vinylene carbonate (VC) as additive. The cyclic voltammetry (CV) profiles were recorded with an electrochemical workstation (PARSTAT2263) at a scan rate of  $0.1 \text{ mV s}^{-1}$  in a potential window of 0.01–3.0 V. Electrochemical impedance spectroscopy (EIS) measurements were carried out on an electrochemical workstation (PARSTAT2263). Galvanostatic discharge/charge tests were conducted using a Land-CT2001A battery cycler (Wuhan, China) at room temperature.

## 3. RESULTS AND DISCUSSION

**3.1. Structure and Morphology Characterization.** In the synthesis process of adding PVP K-30, the hierarchical microspheres of Mn-EG precursors were first prepared and confirmed by XRD and IR characterization (Supporting Information Figure S1).<sup>15,37</sup> The hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres were obtained after the calcination of Mn-EG precursors. The XRD spectra of as-prepared  $\text{Mn}_2\text{O}_3$  are shown in Figure 1. All the diffraction peaks can be perfectly indexed to the cubic bixbyite  $\alpha\text{-Mn}_2\text{O}_3$  (JCPDS, No. 41-1442). No peaks corresponding to other phases are found, confirming the high purity of the synthesized sample.

The SEM image in Figure 2 reveals that the resulting  $\text{Mn}_2\text{O}_3$  hierarchical hollow microspheres are constituted of porous microspheres (Figure 2e). The diameter of the  $\text{Mn}_2\text{O}_3$  microspheres is 6–10  $\mu\text{m}$ , which is in accordance with the size of the commercial LIBs materials. And the  $\text{Mn}_2\text{O}_3$  microspheres have a same size as that of the Mn-EG precursors (Figure 2b), but present a porous structure after calcination. The thickness of the porous microspheres is measured as 100 nm (Figure 2f), which is about 80 nm thicker than the Mn-EG precursors due to agglomeration of neighboring microspheres of Mn-EG precursors in calcination process (Figure 2c). However,



**Figure 1.** XRD patterns of  $\text{Mn}_2\text{O}_3$  JCPDS standard (card number 42-0580) and  $\text{Mn}_2\text{O}_3$  prepared by solvent-thermal method.

the product without adding PVP K-30 in the synthesis process presents microspheres structure shown in Figure S2. The specific surface area of the as-prepared  $\text{Mn}_2\text{O}_3$  hierarchical hollow microspheres was measured at  $7.83 \text{ m}^2 \text{ g}^{-1}$  by BET using  $\text{N}_2$  absorption (Figure S3).

Figure 3 depicts the detailed structural features of  $\text{Mn}_2\text{O}_3$  microspheres. From TEM and HRTEM images, we observe clearly hollow structure and atomic lattice of the  $\text{Mn}_2\text{O}_3$ . The hollow construction features illustrated in Figure 3a may ensure that the  $\text{Mn}_2\text{O}_3$  microspheres can mitigate the local volume variation during charge–discharge cycling. It is interesting to see from Figure 3b that the particles are closely linked to form the porous microspheres, which is electric conductive and guarantees the structural stability in conversion reaction. High-resolution TEM (HRTEM) image displayed in Figure 3c indicates the lattice fringe with an interplanar distance of ca.  $2.68 \text{ \AA}$ , corresponding to the (222) plane of cubic  $\text{Mn}_2\text{O}_3$ .

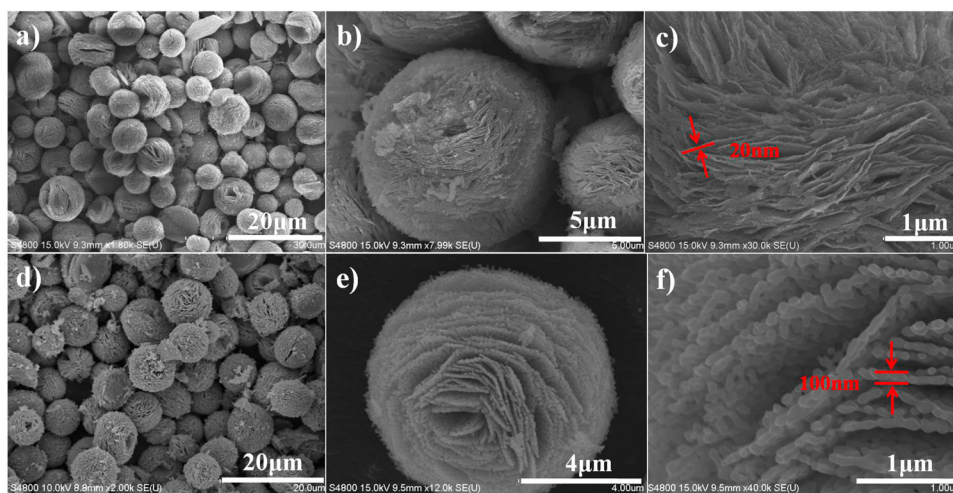
**3.2. Electrochemical Performances of the  $\text{Mn}_2\text{O}_3$  Hierarchical Hollow Microspheres.** The electrochemical performances of the as-synthesized  $\text{Mn}_2\text{O}_3$  were investigated by cyclic voltammetry (CV) and charge–discharge cycling (Figure 4). CV curves of the first three cycles of the  $\text{Mn}_2\text{O}_3$  electrode are shown in Figure 4a. Two reductive peaks located at 1.1 and 0.2 V are clearly observed in the first cathodic process. The broad peak at about 1.1 V can be ascribed to the reduction of  $\text{Mn}^{3+}$  to  $\text{Mn}^{2+}$  and an irreversible reaction related to the formation of SEI layer.<sup>32</sup> The reduction action of  $\text{Mn}^{2+}$  to metallic Mn occurs near 0.2 V. However, only one peak is

observed at 1.2 V in the first anodic process, indicating the lithium ion extraction reaction of  $\text{Li}_2\text{O}$  phase.<sup>16,30,38</sup> After the first cycle, nevertheless, only one cathodic peak at about 0.25 V and one anodic peak at 1.2 V are observed.

Figure 4b presents the first two discharge–charge curves of the porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres electrode at a current density of  $500 \text{ mA g}^{-1}$ . The first discharge profile consists of a slope at about 1.2 V and a wide plateau near 0.2 V, which consume an overall capacity of  $1210 \text{ mAh g}^{-1}$ . In the first charge curves, a wide slope appears at 1.2 V and the capacity is  $760 \text{ mAh g}^{-1}$  with a Coulombic efficiency of about 62.8%. The discharge plateau is shifted to about 0.5 V in the second cycle, and no change for charge plateau.

Figure 4c and d demonstrates the cycling performance and rate capability of the  $\text{Mn}_2\text{O}_3$  electrode in galvanostatic experiments. A reversible capacity of  $580 \text{ mAh g}^{-1}$  at  $500 \text{ mA g}^{-1}$  with a high Coulombic efficiency about 99.7% was measured after 140 cycles (Figure 4c), maintaining higher capacity and retention rate than  $\text{Mn}_2\text{O}_3$  microspheres (Figure S4). To evaluate the rate capability, the  $\text{Mn}_2\text{O}_3$  electrode was charge–discharged at different rates. As illustrated in Figure 4d, reversible charge capacities of 751, 611, 511, and  $422 \text{ mAh g}^{-1}$  were measured respectively at 100, 400, 800, and  $1600 \text{ mA g}^{-1}$ . Compared to the data reported recently,<sup>32,33</sup> our  $\text{Mn}_2\text{O}_3$  microspheres possess excellent electrochemical performance, which is attributed to (a) the porous hollow structural character ensuring the structure stability during conversion cycling, (b) the hierarchical structure making each nanosheet contact directly with electrolyte, and (c) the pores between  $\text{Mn}_2\text{O}_3$  nanoparticles allowing the fast diffusion of  $\text{Li}^+$ . In addition, the  $\text{Mn}_2\text{O}_3$ /carbon/LA133 electrodes after 140 cycles at  $500 \text{ mA g}^{-1}$  on copper foil were characterized by SEM as shown in Figure S5. Some of the porous and hierarchical structure still is preserved after such a long cycling process, indicating high structure stability.

**3.3. XANES Study of Conversion Reaction Mechanism of  $\text{Mn}_2\text{O}_3$  Electrode Materials.** The XANES spectra of reference compounds of Mn, MnO,  $\text{Mn}_3\text{O}_4$ , and  $\text{Mn}_2\text{O}_3$  (Figure S6) display an absorption Mn K-edge at 6539.0, 6544.2, 6547.0, and 6548.2 eV, respectively. The XANES spectra of the MnO,  $\text{Mn}_3\text{O}_4$ , and  $\text{Mn}_2\text{O}_3$  exhibit a small pre-edge peak around 6539 eV. Due to the hybridization of the p



**Figure 2.** SEM images of Mn-ethylene glycol (EG) precursors (a, b, c) and the porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres (d, e, f) at different magnifications.

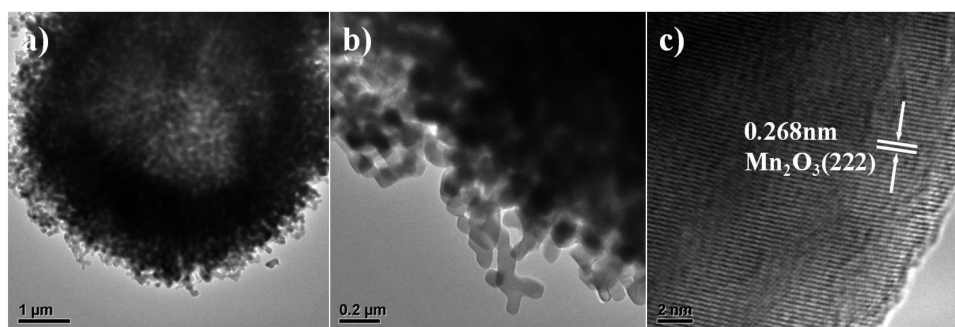


Figure 3. TEM (a, b) and HRTEM (c) images of the porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres.

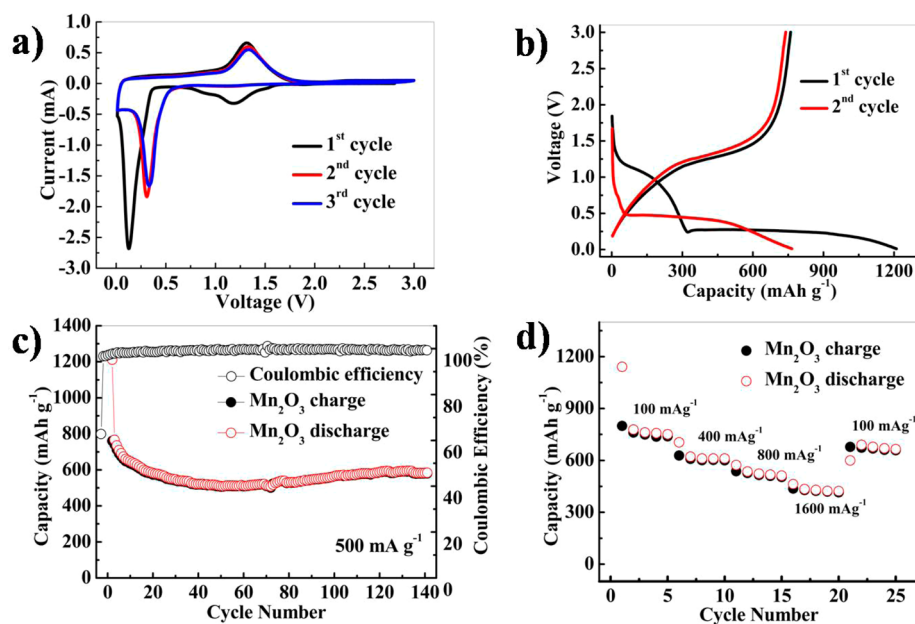


Figure 4. Electrochemical properties of the porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres anode: (a) first three consecutive CVs, (b) discharge–charge curves at a current of  $500 \text{ mA g}^{-1}$ , (c) cycling performance of the porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres at a current of  $500 \text{ mA g}^{-1}$ , and (d) rate capability of the porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres.

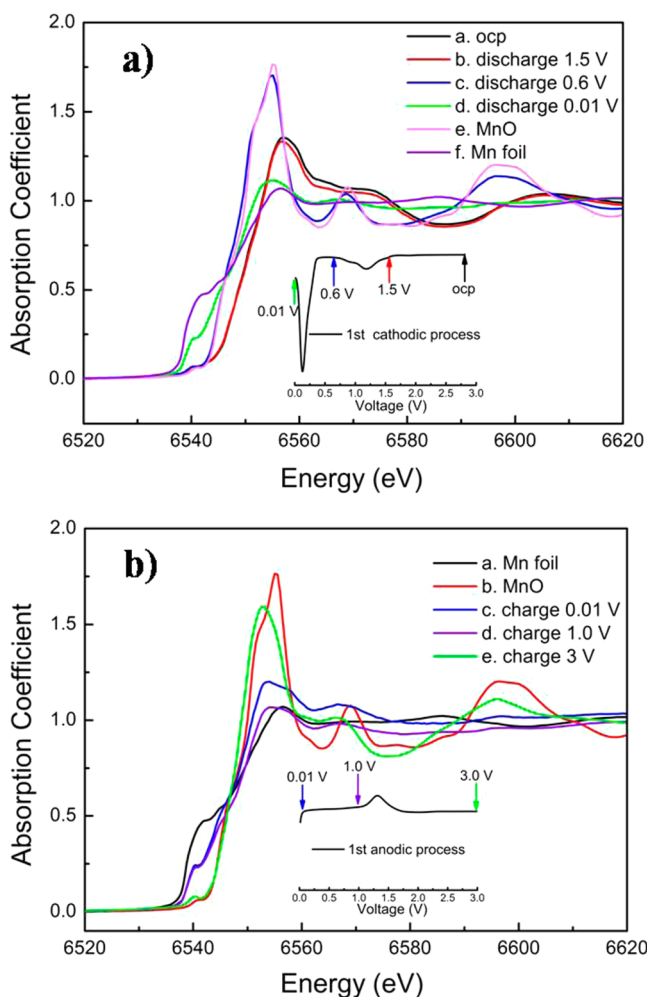
and d orbitals, the transition from 1s to 3d is normally forbidden, which is different from the case of the metallic Mn K-edge.<sup>39</sup>

The K-edge evolution of Mn element in the as-synthesized  $\text{Mn}_2\text{O}_3$  microspheres in the first cycle was studied by ex situ XANES. As illustrated in Figure 5, when  $\text{Mn}_2\text{O}_3$  is discharged to 1.5 V (Figure 5a), the  $E_0$  (6548.2 eV) of the Mn K-edge is the same as that measured at open circuit potential (OCP), indicating no valence reduction in this stage, which is consistent with the electrochemical measurements. Compared to the reference compounds, the  $E_0$  shifts to low energy and is the same as that of MnO when the  $\text{Mn}_2\text{O}_3$  is discharged to 0.6 V, indicating that  $\text{Mn}^{3+}$  is reduced to  $\text{Mn}^{2+}$ . At the end of discharge process, the pre-edge peak disappears and the Mn K-edge position shifts to lower energy, illustrating a further reduction of the  $\text{Mn}^{2+}$  to  $\text{Mn}^0$ .

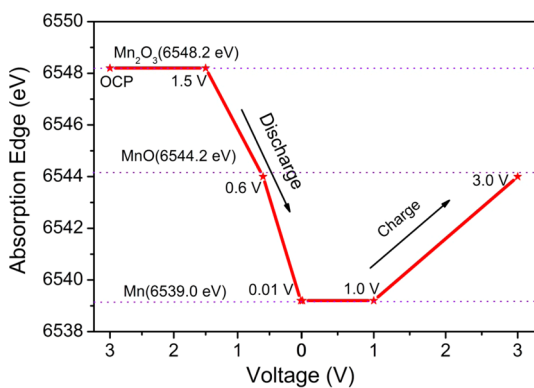
During the charge process,  $E_0$  shifts to the inverse trend, which can be interpreted as an increase in the valence of the Mn (Figure 5b). When charged from 0.01 to 1.0 V, no obvious shift in the K-edge of Mn could be observed in comparison with that at 0.01 V, illustrating that the valence of Mn remains unchanged and the conversion reaction has not occurred. From 1.0 to 3 V in the charge process, the pre-edge peak reappears

and the  $E_0$  is the same as that of MnO. As illustrated in Figure 6, the  $E_0$  at different states measured at characteristic points is shifted to low/high energy in discharge/charge processes, demonstrating clearly the valence evolution of Mn during the electrochemical conversion reaction processes.

To investigate the origin of the capacity loss of  $\text{Mn}_2\text{O}_3$  materials during cycling, we analyzed the Mn K-edge XANES spectra of the  $\text{Mn}_2\text{O}_3$  sample subjected to different cycling numbers (Figure 7). In the first charge/discharge cycle, the pristine  $\text{Mn}_2\text{O}_3$  is reduced to metallic Mn when discharged to 0.01 V and oxidized to MnO while charged to 3.0 V. However, for the  $\text{Mn}_2\text{O}_3$  samples subjected to 5, 10, and 50 cycles, the Mn K-edge energy shifts to low energy for the delithiated state (i.e., 6543.5, 6543.3, and 6543.6 eV, respectively) and keeps stable for the lithiated state (about 6539 eV). This indicates that the conversion reaction of metallic Mn and  $\text{Li}_2\text{O}$  phase to oxide (mainly MnO) is not a fully reversible reaction. As a consequence, the valence of Mn in the sample at charged state has decreased along with cycling, which leads to reduction of the amount of electrons involved in the reaction. As a result, the accumulation of  $\text{Li}_2\text{O}$  which does not participate in deconversion reaction makes the charge transfer resistance ( $R_{ct}$ ) increase (Figure S7) and the capacity decay.

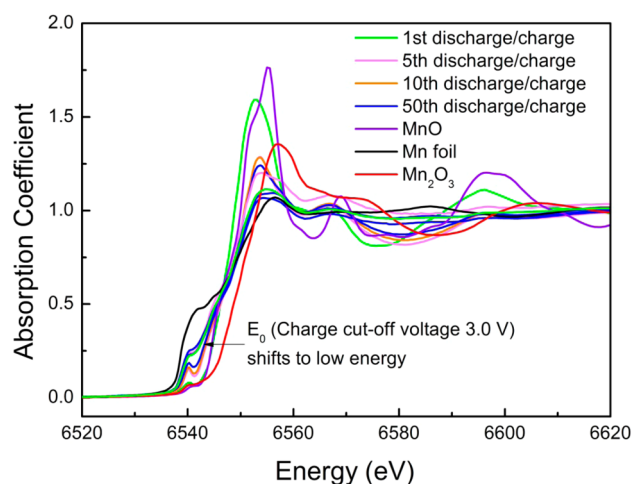


**Figure 5.** Ex situ Mn K-edge XANES spectra of  $\text{Mn}_2\text{O}_3$  electrode at different preset voltages during the first discharging (a)/charging (b) process and Mn K-edge XANES spectra of Mn, MnO, and  $\text{Mn}_3\text{O}_4$ .



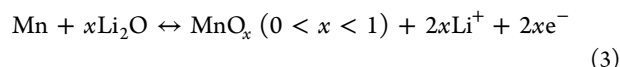
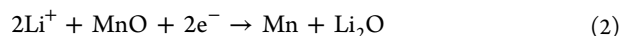
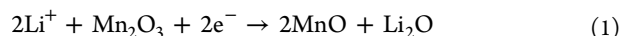
**Figure 6.**  $E_0$  evolution of the Mn K-edge along with voltage change during the first discharge and charge processes.

From above ex situ XANES results, the conversion reaction processes of  $\text{Mn}_2\text{O}_3$  electrodes may be suggested as follows. During the first discharge, the slope from 1.5 to 0.7 V may associate with the reduction of  $\text{Mn}^{3+}$  to  $\text{Mn}^{2+}$ , as expressed by eq 1 below. Then, a broad plateau appears at about 0.2 V, indicating the reduction action of  $\text{Mn}^{2+}$  to metallic Mn, as expressed by eq 2. From the first charge process, the



**Figure 7.** Ex situ Mn K-edge XANES spectra of different cyclic numbers  $\text{Mn}_2\text{O}_3$  electrodes and Mn K-edge XANES spectra of Mn and MnO.

electrochemical reversible reaction undergoes the conversion between Mn and  $\text{MnO}_x$ , corresponding to eq 3.



#### 4. CONCLUSION

In the current study, a solvent-thermal method was used to synthesize porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres, which served as anode materials for the lithium ion battery. These special structural features of  $\text{Mn}_2\text{O}_3$  guarantee its excellent electrochemical performance during charge/discharge processes. The hierarchical structure can ensure every subunit contacting with the electrolyte adequately, the pores between nanoparticles can facilitate both  $\text{Li}^+$  and electron diffusion, and the hollow structure can mitigate efficiently the local volume variation. The as-synthesized porous hierarchical  $\text{Mn}_2\text{O}_3$  hollow microspheres anode can deliver a reversible capacity of  $580 \text{ mAh g}^{-1}$  at a current density of  $500 \text{ mA g}^{-1}$  with a high coulombic efficiency about 99.7% after 140 cycles of charge–discharge. At high current densities of charge–discharge of 100, 400, 800, and  $1600 \text{ mA g}^{-1}$ , the  $\text{Mn}_2\text{O}_3$  electrodes can still deliver reversible charge capacities of 751, 611, 511, and  $422 \text{ mAh g}^{-1}$ , respectively. The XANES spectra were used to characterize the Mn K-edge, and the results demonstrate that the  $\text{Mn}_2\text{O}_3$  is reduced to metallic Mn when discharged to 0.01 V and oxidized to  $\text{Mn}^{2+}$  while charged to 3 V in the first cycle. In the subsequent cycles of charge–discharge, the average valence of Mn in the sample at charged state decreases, resulting from reducing the amount of electrons participating in the reaction, which is the origin of the irreversible capacity loss of  $\text{Mn}_2\text{O}_3$  materials.

#### ■ ASSOCIATED CONTENT

##### Supporting Information

XRD pattern and FT-IR spectrum of the Mn-EG, SEM images of the  $\text{Mn}_2\text{O}_3$  microspheres without PVP K-30 in the synthesis process,  $\text{N}_2$  adsorption/desorption measurements of the porous

hierarchical Mn<sub>2</sub>O<sub>3</sub> hollow microspheres, cycling performance of the Mn<sub>2</sub>O<sub>3</sub> microsheets at a current of 500 mA g<sup>-1</sup> in a potential window of 0.01–3.0 V, SEM images of the Mn<sub>2</sub>O<sub>3</sub>/carbon/LA 133 electrodes after 140 cycles at 500 mA g<sup>-1</sup> on copper foil, reference compounds' Mn K-edge X-ray absorption near-edge structure (XANES), and EIS spectra of different cyclic numbers Mn<sub>2</sub>O<sub>3</sub> electrodes. This material is available free of charge via the Internet at <http://pubs.acs.org/>.

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### Notes

The authors declare no competing financial interest.

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